Drastic Enhancement of the $CO₂$ Adsorption Properties in Sulfone-Functionalized Zr- and Hf-UiO-67 MOFs with Hierarchical Mesopores

Pantelis Xydias, Ioannis Spanopoulos, Emmanuel Klontzas, George E. Froudakis, and Pantelis N. Trikalitis*

Department of Chemistry, Uni[ve](#page-2-0)rsity of Crete, Voutes 71003, Heraklion, Greece

S Supporting Information

[AB](#page-2-0)STRACT: [The](#page-2-0) [sulfone-](#page-2-0)functionalized Zr- and Hf-UiO-67 metal−organic frameworks with hierarchical mesopores were successfully synthesized using the ligand 4,4′-dibenzoic acid-2,2′-sulfone, with acetic acid or HCl as the modulator. Compared to UiO-67, the zirconium solid shows a remarkable 122% increase in $CO₂$ uptake, reaching 4.8 mmol g^{-1} (17.4 wt %) at 1 bar and 273 K (145% at 298 K) and more than 100% increase in $CO₂/$ CH₄ selectivity.

I urrently, an enormous research activity is devoted to porous metal−organic frameworks (MOFs) or porous coordination polymers mainly because of the prospect of finding application in important technological sectors related to energy and the environment.¹ In particular, the reduction of greenhouse gas emissions and especially $CO₂$ is currently a major target worldwide, and high[ly](#page-2-0) porous sorbents like MOFs hold great promise as cost-effective alternatives to the existing technologies.1b,2,3

A particular MOF that has captured a great deal of attention bec[ause](#page-2-0) of its high thermal and chemical stability is the zirconium-based, 12-connected $Zr_6(\mu_3$ -O)₄(μ_3 -OH)₄(bdc)₆ (UiO-66) microporous material, made with terephthalate dianions ($bdc^{2−}$).⁴ This material shows interesting CO_2 sorption properties especially for low-pressure and ambient-temperature applications that [c](#page-2-0)an be further improved with functionalized $H₂$ bdc linkers⁵ and, in particular, with those containing the highly polar sulfonic acid group, as has been demonstrated both theoretically^{5[b](#page-2-0)} and experimentally.^{5c} The isostructural analogue UiO-67, based on the linker 4,4′-biphenyldicarboxylate (bpdc²[−]), b[eca](#page-2-0)use of its larger [por](#page-2-0)e size and increased pore volume, is considered as a very promising material for $CO₂$ adsorption. 6 However, related studies are limited, 7 and to the best of our knowledge, there is no report on functionalized UiO-67.

We report here the synthesis and characterization of the sulfone-functionalized analogue of $UiO-67(Zr)$, using the ligand 4,4'-dibenzoic acid-2,2'-sulfone (H_2bbs) , in the presence of acetic acid (AcOH) or HCl as the modulator, denoted as $1_{A_{COH}}$ and 1_{HC} , respectively. We have also isolated the hafnium analogues, denoted as $2_{\rm AcOH}$ and $2_{\rm HCl}$. For comparison purposes, we synthesized Zr-UiO-67 using the ligand 4,4′-biphenyldicarboxylic acid (H₂bpdc), denoted as $3_{A_{COH}}$ and 3_{HCl} . Remarkably, compared to the parent, high-quality nonfunctionalized solid 3_{AcOH} , 1_{AcOH} shows a 122% increase in CO_2 uptake at 1 bar and

273 K (145% at 298 K), reaching 4.8 mmol g^{-1} (2.88 mmol g^{-1} at 298 K), with high selectivity toward CH_4 and N_2 . In terms of key structural features, we provide a direct proof, using ¹H NMR spectroscopy, for the presence of AcO[−] anions coordinated to the $\rm Zr_6$ clusters in $\rm 1_{\rm AcOH}$. Moreover, both AcOH and HCl induce in $1_{\text{A}\text{cOH}}$ and 1_{HCl} hierarchical mesopores. Below, we present and discuss a detailed structural and gas sorption study of the zirconium-based solids, although key findings for the hafnium analogues are also reported.

Because of the lack of suitable single crystals, the ideal structure of 1 shown in Figure S2 was built using a ligand replacement methodology starting from the known structure of UiO-66.⁴ The powder X-ray diff[rac](#page-2-0)tion (PXRD) patterns of high-quality $1_{A\text{cOH}}$ and $2_{A\text{cOH}}$ are shown in Figure 1. All

Figure 1. Powder X-ray diffraction pattern of 1_{AcOH} (left) and 2_{AcOH} (right). Insets: representative SEM images.

compounds are isostructural, and the PXRD pattern can be indexed to the cubic system with refined unit cells of $26.61(2)$, 26.54(3), and 26.87(2) Å for 1_{AcOH} , 2_{AcOH} , and 3_{HCl} , respectively. The smaller unit cell of $1_{A\text{coH}}$ and $2_{A\text{coH}}$ compared to 3_{HCl} is due to the shorter length of the bbs^{2−} linker caused by the bending of the two aromatic rings. It is worth noticing that, despite its large deviation from linearity (163°), this linker is capable of forming an isostructural solid with UiO-67.

Argon sorption isotherms at 87 K for $\mathbf{1}_{\text{ACOH}}$ $\mathbf{1}_{\text{HCb}}$ and $\mathbf{3}_{\text{HCl}}$ are shown in Figure 2, from which important porosity data were determined and are summarized in Table S1 in the Supporting Information (SI). [A](#page-1-0)ccordingly, both $\mathbf{1}_{A\text{cOH}}$ and $\mathbf{1}_{HCl}$ show almost identical Brunauer–Emmett–Teller (BET) areas, 1442 m² g⁻¹ [\(Langmuir 1](#page-2-0)597 m² g⁻¹) and 1456 m² g⁻¹ (Langm[uir](#page-2-0) [1601](#page-2-0) m² $\rm g^{-1})$, respectively, which are lower compared to $\rm 3_{HCl}$ (BET area 1996 $\mathrm{m^{2} g^{-1}}$ and Langmuir 2253 $\mathrm{m^{2} g^{-1}}$) presumably because of

Received: September 25, 2013 Published: December 24, 2013

Figure 2. Argon adsorption (closed circles) and desorption (open circles) isotherms at 87 K of $\mathbf{1}_{\text{ACOH}}$ (blue), $\mathbf{1}_{\text{HCl}}$ (red), and $\mathbf{3}_{\text{HCl}}$ (green). Inset: hysteresis in $\mathbf{1}_{\text{ACOH}}$ and $\mathbf{1}_{\text{HCl}}$

the presence of the sulfone groups. The PXRD patterns of the activated samples 1_{AcOH} , 2_{AcOH} , 3_{HC} , and 3_{AcOH} are identical with those of the as-made materials, while notably for the corresponding 1_{HC} and 2_{HC} , a significantly reduced crystallinity is observed (see the SI). We found that the functionalized solids and especially the evacuated materials, compared to UiO-67, show enhanced moi[stu](#page-2-0)re sensitivity (evacuated solids stored in dry conditions are stable for months), and this is attributed to the presence of the polar sulfone groups that make the solids more hydrophilic. The degradation mechanism of the $Zr_6O_4(OH)_4$ cluster by water in UiO-type solids has been reported recently.⁸ ¹H NMR measurements of acid-digested, activated samples confirm the complete removal of solvent molecules. Moreove[r,](#page-2-0) the $^1\mathrm{H}$ NMR spectrum of $\mathbf{1}_\mathrm{ACOH}$ shows the presence of AcOH in a 1:2.5 ratio with H2bbs. This ratio is reproducible in different batches of activated 1_{AcOH} . Furthermore, energy-dispersive spectroscopy (EDS) shows the absence of Cl[−] anions. These results suggest that the observed AcOH originates from AcO[−] anions coordinated to the Zr_6 clusters at the expense of bridging bbs²[−] linkers. Accordingly, the proposed charge-balanced chemical formula of the activated 1_{ACOH} is $Zr_6(\mu_3$ -O)₄(μ_3 - $\mathrm{OH})_{4}(\mathrm{bbs})_{5}(\mathrm{AcO})_{2}$. This formula is also supported by thermogravimetric analysis (TGA); see the SI. Although in UiO-66 solids it has been recognized that the Zr_6 clusters can be defective, in its AcOH-modulated synthesis, t[he](#page-2-0) presence of a coordinated AcO[−] ligand was confirmed by neutron inelastic scattering experiments very recently, however without quantitative results.⁹ In a more recent report, the presence of coordinated trifluoroacetate anions in UiO-66 was confirmed by solid-state ¹⁹F NMR [s](#page-2-0)pectroscopy.¹⁰ Interestingly, the observed Zr₆ to bdc2[−] ratio was 4.

The large amount of st[oic](#page-2-0)hiometric defects in 1_{AcOH} , due to the presence of coordinated AcO[−] ligands, is expected to create hierarchical mesoporosity. Indeed, accurate pore-size analysis from the argon isotherm of $1_{A_{\rm cOH}}$ confirmed the presence of mesopores, as we discuss below. The presence of a small hysteresis step during desorption at $P/P_0 \sim 0.5$ is in agreement with the existence of mesopores (see the inset in Figure 2). Notably, very similar types of isotherms were observed in MOFs with functionalized mesopores using a metal−ligand-fragment coassembly strategy,¹¹ which, in fact, resembles the synthesis of 1AcOH. Our findings are consistent with a recent report on the formation of meso[po](#page-2-0)res in UiO-66 due to missing linker defects.^{9b}

High-resolution micropore analysis of $1_{A\text{cOH}}$ using argon at 87 K (see [F](#page-2-0)igure S13 in the SI) allowed us to calculate the corresponding pore-size distribution using nonlocal density functional theory (NLDFT). [A](#page-2-0)ccordingly, two distinct peaks centered at 6.3 and 8.4 Å are observed, in agreement with the

expected size of the microporous tetrahedral and octahedral cages in $1_{A_{COH}}$ (see Figure S14 in the SI). In addition, two relatively narrow peaks and one broad, centered at 15, 25, and 50 Å, respectively, are observed, consistent [wi](#page-2-0)th the presence of hierarchical mesopores.

For comparison purposes, we first synthesized and studied the properties of an HCl-modulated UiO-67 (3_{HCl}) because it is reported to be stable.⁶ The corresponding porosity data are summarized in Table S1 in the SI. The argon adsorption isotherm of $3H_{HCD}$ in a[d](#page-2-0)dition to the microporous knee at low relative pressures $(P/P_0 < 0.1)$, shows a well-defined condensation step at $P/P_0 \sim 0.2$, suggesting the presence of uniform, small mesopores. The corresponding NLDFT analysis revealed three distinct peaks centered at 6.4, 9.5, and 27.2 Å (see Figure S34 in the SI). While the first two peaks are expected from the crystal structure (tetrahedral and octahedral cages), the narrow mesopor[ous](#page-2-0) peak at 27.2 Å is observed for the first time in the UiO-67 literature.^{4,6,12} Taking into account that no Cl[−] anions are observed in EDS, TGA suggests the presence of defects, presumably in [the f](#page-2-0)orm of −OH groups coordinated to the Zr_6 clusters. The indicated charge-balanced chemical formula is $\rm{Zr_{6}(\mu_{3}\text{-}O)_{4}(\mu_{3}\text{-}OH)_{4}(bpdc)_{4.7}(OH)_{2.6}}$. Notably, in the HClmodulated synthesis of UiO-66,¹⁰ the Zr_6 :bdc ratio is 5. On the basis of these results, the formation of uniform mesopores in $3H_{HCl}$ is attributed to the large amou[nt](#page-2-0) of HCl used for its synthesis. Compared to the reported pure microporous UiO-67 made with 2 equiv of HCl, $3H_{HCl}$ was made with 35. To verify our proposition, we synthesized 1_{HCl} with a similar large excess of HCl. Indeed, the argon adsorption isotherm at 87 K of 1_{HC} shows a clear condensation step at $P/P_0 \sim 0.1$ consistent with the presence of uniform small mesopores, however larger in size than those in 1_{AcOH} , resulting in an increased total pore volume (see Table S1 in the SI). The observed larger pore size is attributed to the absence of bulky coordinated AcO[−] anions in 1_{HCl} and to additio[na](#page-2-0)l structural defects because of its reduced quality, as judged by scanning electron microscopy (SEM). Accordingly, $1_{\text{A}\text{cOH}}$ consists of uniform particles in single-crystal form (see Figure 1), while highly irregular particles are observed in $1_{\text{HC}1}$. This is also the case between $2_{\text{AcOH}}/2_{\text{HCl}}$ and $3_{\text{AcOH}}/3_{\text{HCl}}$ (see Figure [1](#page-0-0) and the SI), indicating the important role of AcOH in slowing down the reaction kinetics, leading to the formation of uniform well-sha[ped](#page-2-0) particles, in agreement with the literature ${\rm results.}^{12}$ ${\rm results.}^{12}$ ${\rm results.}^{12}$

The CO_2 , CH_4 , N_2 , and H_2 sorption properties were investi[gat](#page-2-0)ed at different temperatures and pressures from which the uptake, isosteric heat of adsorption (Q_{st}) , and selectivity using IAST were calculated and are discussed below (see the SI). Figure 3 shows the low-pressure $CO₂$ adsorption isotherms for $1_{A\text{cOH}}$ and $3_{A\text{cOH}}$ (3_{HCl} performs almost

Figure 3. $CO₂$ adsorption (closed symbols)/desorption (open symbols) isotherms for the indicated solids at 273 K (left) and the corresponding heats of adsorption (right).

identically), at 273 K up to 1 bar. Remarkably, under these conditions, 1_{ACOH} shows a 122% (145% at 298 K) increase in CO_2 uptake compared to $\mathbf{3}_{\mathrm{ACOH}}$, reaching 4.84 mmol g^{-1} , despite its significantly lower BET area (see Table S1 in the SI). This uptake is 20% higher compared to the titanium-substituted Zr-UiO-66^{13a} (4.0 mmol g^{-1}) and among the highest reported in the entire family of UiO-type solids,^{13b} under the same experimental conditions. At 273 K and 20 bar, the CO_2 uptake of $1_{A\text{coH}}$ is 13.3 mmol g $^{-1}$, while at 195 K and 1 bar, it is 16.5 mmol $\rm g^{-1}$ (see the SI). With an identical BET area, 1_{HC} at 273 K and 1 bar adsorbs 4.0 mmol $\rm g^{-1}$. In comparison to $\rm 1_{AcOH}$ the 17% decrease in $\rm CO_2$ uptake is related to the reduced number of bbs^{2−} linkers (less sulfone groups) per Zr_6 cluster in 1_{HCl} (4.7 versus 5) and also to the increased pore size (reduced attracting potential). In line with that above, the heavier hafnium analogues, 2_{HCl} and $2_{\text{A}\text{cOH}}$, adsorb 3.5 and 4.5 mmol g^{-1} of CO₂ at 273 K and 1 bar, respectively (see Table S1 in the SI and Figure 3).

The calculated Q_{st} values at zero coverage (Q_{st}°) using the adsorption data at 273 and 298 K are 26.5 and [18](#page-1-0).5 kJ mol[−]¹ for 1_{AcoH} and 3_{AcoH} (16.9 kJ mol⁻¹ for 3_{HCl}), respectively. As a function of the surface coverage, Q_{st} is slightly decreasing for 1_{AcOH} , reaching 22.9 kJ mol⁻¹ at high coverage (see Figure 3). These medium-ranged Q_{st} values are considered promising for low-cost regeneration of porous CO_2 adsorbers.^{1b,2b,3} [Th](#page-1-0)e substantial increase of 8 kJ mol⁻¹ in Q_{st}° between $1_{A\text{cOH}}$ and 3_{AcOH} is attributed mainly to the polar sulfone groups that interact stronger with $CO₂$ because of its quadrupole moment.^{1b} Accurate ab initio calculations show that, in going from the H_2 bpdc to the H_2 bbs linker, the increase in the CO_2 binding strength is approximately 30% (3.2 and 4.1 kcal mol⁻¹, , respectively). However, for the observed increase in the $CO₂$ uptake (145% at 298 K) and 55% in $Q_{\rm st}$, additional factors are contributing, including a decrease in the pore size, which increases the overlap of attractive potential fields of opposite surfaces, leading to stronger and higher $CO₂$ adsorption.

Regarding CH₄, Q_{st} ° values for $1_{A\text{coH}}$ and 3_{HCl} are 14.5 and 8.9 kJ mol^{−1}, respectively. The $\rm CO_2/CH_4$ selectivity in $\rm\,1_{\rm\Lambda cOH}$ at 273 K/298 K using IAST for a 5/95 molar mixture is 9.8/6.8, while for $3H\text{HCD}$ it is significantly lower, 4.0/3.2 (see the SI). The observed more than 100% increase in selectivity is attributed to the increased interaction of $CO₂$ with the polar sulfone groups. The CO_2/N_2 selectivity in $1_{A\text{cOH}}$ at 273 K/298 K is 23.9/24.5. The $\rm CO_2/CH_4$ and $\rm CO_2/N_2$ selectivities in $\rm\,1_{ACOH}$ are among the highest for MOFs without open metal sites or amine groups.^{1b} Finally, the H₂ uptake at 77 K is 25% higher in $1_{A_{COH}}(1.5 \text{ wt } \%)$ compared to $3HCl$ (1.2 wt %), with Q_{st}° values of 7.4 and 6.3 kJ mol^{−1}, respectively.

In conclusion, we have shown that the insertion of polar sulfone groups into UiO-67 increases drastically the $CO₂$ uptake and selectivity. Control over missing linker defects using AcOH and HCl or other types of modulators offers the possibility of preparing materials with adjustable hierarchical mesoporosity. In addition, given the available organic linkers that can be functionalized with sulfone groups, our findings could be utilized to prepare a family of new zirconium-based MOFs with improved $CO₂$ adsorption properties.

ASSOCIATED CONTENT

6 Supporting Information

Experimental procedures, gas sorption isotherms and related calculations, IR, NMR, TGA, and SEM/EDS. This material is available free of charge via the Internet at http://pubs.acs.org.

■ AUTHOR INFORMATION

Corresponding Author

*E-mail: ptrikal@chemistry.uoc.gr.

Notes

The aut[hors declare no competing](mailto:ptrikal@chemistry.uoc.gr) financial interest.

■ ACKNOWLEDGMENTS

This work was supported by the "Heracleitus II" and "THALES" programs funded by GSRT in Greece.

■ REFERENCES

(1) (a) Suh, M. P.; Park, H. J.; Prasad, T. K.; Lim, D.-W. Chem. Rev. 2012, 112, 782−835. (b) Sumida, K.; Rogow, D. L.; Mason, J. A.; McDonald, T. M.; Bloch, E. D.; Herm, Z. R.; Bae, T.-H.; Long, J. R. Chem. Rev. 2012, 112, 724−781. (c) Makal, T. A.; Li, J. R.; Lu, W. G.; Zhou, H. C. Chem. Soc. Rev. 2012, 41, 7761−7779.

(2) (a) Markewitz, P.; Kuckshinrichs, W.; Leitner, W.; Linssen, J.; Zapp, P.; Bongartz, R.; Schreiber, A.; Muller, T. E. Energy Environ. Sci. 2012, 5, 7281−7305. (b) Li, J. R.; Ma, Y. G.; McCarthy, M. C.; Sculley, J.; Yu, J. M.; Jeong, H. K.; Balbuena, P. B.; Zhou, H. C. Coord. Chem. Rev. 2011, 255, 1791−1840.

(3) (a) Nugent, P.; Belmabkhout, Y.; Burd, S. D.; Cairns, A. J.; Luebke, R.; Forrest, K.; Pham, T.; Ma, S. Q.; Space, B.; Wojtas, L.; Eddaoudi, M.; Zaworotko, M. J. Nature 2013, 495, 80−84. (b) Xue, D. X.; Cairns, A. J.; Belmabkhout, Y.; Wojtas, L.; Liu, Y.; Alkordi, M. H.; Eddaoudi, M. J. Am. Chem. Soc. 2013, 135, 7660−7667.

(4) Cavka, J. H.; Jakobsen, S.; Olsbye, U.; Guillou, N.; Lamberti, C.; Bordiga, S.; Lillerud, K. P. J. Am. Chem. Soc. 2008, 130, 13850−13851. (5) (a) Biswas, S.; Van der Voort, P. Eur. J. Inorg. Chem. 2013, 2154− 2160. (b) Yang, Q. Y.; Wiersum, A. D.; Llewellyn, P. L.; Guillerm, V.; Serred, C.; Maurin, G. Chem. Commun. 2011, 47, 9603−9605. (c) Biswas, S.; Zhang, J.; Li, Z. B.; Liu, Y. Y.; Grzywa, M.; Sun, L. X.; Volkmer, D.; Van der Voort, P. Dalton Trans. 2013, 42, 4730−4737. (d) Jasuja, H.; Zang, J.; Sholl, D. S.; Walton, K. S. J. Phys. Chem. C 2012, 116, 23526−23532. (e) Foo, M. L.; Horike, S.; Fukushima, T.; Hijikata, Y.; Kubota, Y.; Takata, M.; Kitagawa, S. Dalton Trans. 2012, 41, 13791− 13794. (f) Kim, M.; Cohen, S. M. CrystEngComm 2012, 14, 4096−4104. (g) Abid, H. R.; Ang, H. M.; Wang, S. B. Nanoscale 2012, 4, 3089−3094. (6) Yang, Q. Y.; Guillerm, V.; Ragon, F.; Wiersum, A. D.; Llewellyn, P. L.; Zhong, C. L.; Devic, T.; Serre, C.; Maurin, G. Chem. Commun. 2012, 48, 9831−9833.

(7) Nik, O.; Chen, X. Y.; Kaliaguine, S. J. Membr. Sci. 2012, 413, 48−61. (8) DeCoste, J. B.; Peterson, G. W.; Jasuja, H.; Glover, T. G.; Huang, Y. G.; Walton, K. S. J. Mater. Chem. A 2013, 1, 5642−5650.

(9) (a) Valenzano, L.; Civalleri, B.; Chavan, S.; Bordiga, S.; Nilsen, M. H.; Jakobsen, S.; Lillerud, K. P.; Lamberti, C. Chem. Mater. 2011, 23, 1700−1718. (b) Wu, H.; Chua, Y. S.; Krungleviciute, V.; Tyagi, M.; Chen, P.; Yildirim, T.; Zhou, W. J. Am. Chem. Soc. 2013, 135, 10525− 10532.

(10) Vermoortele, F.; Bueken, B.; Le Bars, G. L.; Van de Voorde, B.; Vandichel, M.; Houthoofd, K.; Vimont, A.; Daturi, M.; Waroquier, M.; Van Speybroeck, V.; Kirschhock, C.; De Vos, D. E. J. Am. Chem. Soc. 2013, 135, 11465−11468.

(11) Park, J.; Wang, Z. Y. U.; Sun, L. B.; Chen, Y. P.; Zhou, H. C. J. Am. Chem. Soc. 2012, 134, 20110−20116.

(12) Schaate, A.; Roy, P.; Godt, A.; Lippke, J.; Waltz, F.; Wiebcke, M.; Behrens, P. Chem.-Eur. J. 2011, 17, 6643-6651.

(13) (a) Lau, C. H.; Babarao, R.; Hill, M. R. Chem. Commun. 2013, 49, 3634−3636. (b) Jiang, H. L.; Feng, D. W.; Liu, T. F.; Li, J. R.; Zhou, H. C. J. Am. Chem. Soc. 2012, 134, 14690−14693.